



## **Title: BFO films obtained by Spray Pyrolysis optical and structural analysis**

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**Editorial label ECORFAN: 607-8695**

**BCIERMMI Control Number: 2022-01**

**BCIERMMI Classification (2022): 261022-0001**

**Pages: 15**

**RNA: 03-2010-032610115700-14**

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# Introduction

As well known, multiferroics are interesting materials which exhibits at least two of the ferroic properties in the same phase and exhibit a coupling effect between the ferroic properties [1].

Among many ferroelectric materials, BiFeO<sub>3</sub> (BFO) is one of the key research materials due to its large remanent polarization ( $\sim 100 \mu\text{C}/\text{cm}^2$ ), high Curie temperature ( $\sim 810^\circ\text{C}$ ) and narrow direct band gap ( $\sim 2.7 \text{ eV}$ ) [2], whereby it has significant absorption of visible light, which is beneficial for the development of solar energy devices [3]. Nevertheless, the major challenge on BiFeO<sub>3</sub> based photovoltaic devices is low photovoltaic output, which is affected by their intrinsic ferroelectric photovoltaic mechanism [4].

Besides to photovoltaic applications, the BFO has also applications in the field of spintronics, sensors and new data storage devices. [5].

Many physical and chemical methods have been used to obtain BFO films, including radio-frequency sputtering, molecular beam epitaxy, pulsed laser deposition, chemical (sol–gel) solution deposition, and hydrothermal synthesis, have been employed for the deposition of BFO thin films. Notwithstanding, these deposition techniques are highly expensive, and it is required long time for deposition process. On the other hand, spray pyrolysis is a very simple and inexpensive deposition technique for thin films fabrication, and it has potential for large scale preparation. Additionally, precise control of composition and better chemical homogeneity are possible in this deposition technique [6].

The growing condition which allows a pure and stoichiometric BFO phase to be obtained is always very narrow, so that very small changes in the deposition parameters can lead to either major changes in the film's properties or to the nucleation of spurious phases. The BFO films have been reported to present a magnetic moment arising from parasitic phases formed mainly by iron oxides [7].

In the case of impurity phases that present an excess of bismuth, such as Bi<sub>25</sub>FeO<sub>39</sub> or the combination of BFO with B<sub>2</sub>O<sub>3</sub>, a great interest has recently arisen in their formation, due to investigations that report the superior photocatalytic action that they exhibit [8] along with good photoelectric response.

# Description of the method

The BFO films were deposited using the Spray Pyrolysis technique, the deposit temperature used was 100°C and for preparing the precursor solution a molarity of 0.6 M was used, it was required as a first step to use a reaction for which the following reagents were used; Iron nitrate nonahydrate 98% and Bismuth Nitrate pentahydrate

Causing the following reaction:



The deposition time used was 10 minutes and the deposition were carried out on an n-type silicon substrate with a resistivity of 1-10  $\Omega$ , orientation (1 0 0), the deposit distance between the hot wall and the substrate was 4 cm.

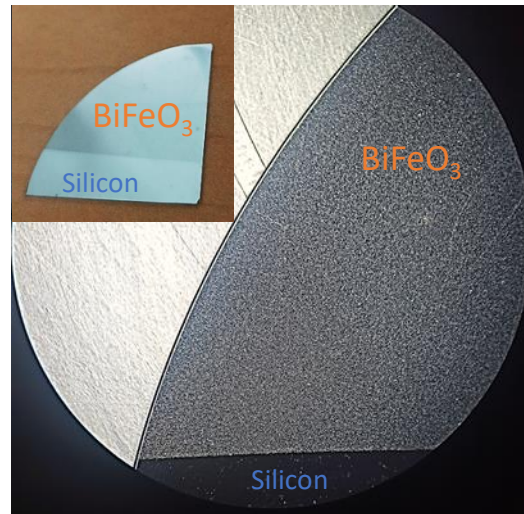
After the deposit, a thermal annealing of 500°C was carried out in accordance with what was reported by [9], in that investigation annealing at higher temperatures was carried out, resulting in the emergence of phases Bi<sub>2</sub>Fe<sub>4</sub>O<sub>9</sub> and Bi<sub>25</sub>FeO<sub>39</sub> at 700°C temperature and at temperatures higher than 800°C there is a decrease in the phase Bi<sub>25</sub>FeO<sub>39</sub> and the phase Bi<sub>2</sub>Fe<sub>4</sub>O<sub>9</sub> becomes dominant.

To structurally characterize the obtained film, profilometry measurements were made in a Dektak 150 profilometer with a vertical tracking resolution of 1 Å, SEM measurements were also made with a JEOL JSM-5300 equipment applying 20kV. Finally, grazing incidence X-ray diffraction measurements, with an angle  $\theta$ -2 $\theta$  of 1° using a Bruker D8 Venture equipment was carried out.

To obtain the optical properties of the material, diffuse reflectance and absorbance measurements were made using a Varian Cary 5000 UV-Vis-NIR spectrophotometer. Finally, the obtained structures were modeled using the Diamond 4 Cristal and molecular structure visualization software.

# Results

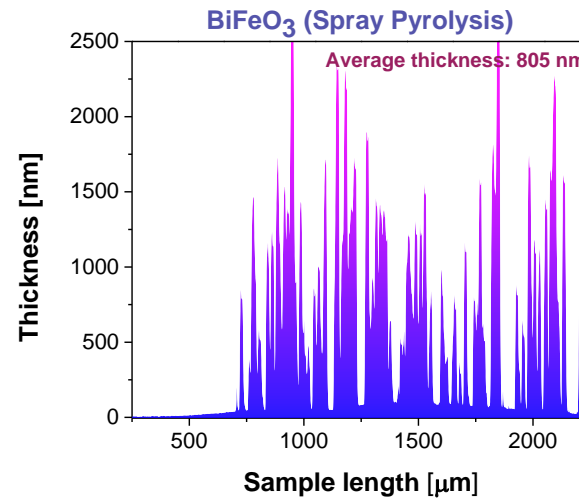
In Figure 1, a photograph of the appearance of the film obtained without close-ups is presented in the upper left part, while in the lower part a photograph of the film taken with microscope magnification is shown with the presence of small clusters within the film.



**Figure 1** *Image of the film obtained using the spray pyrolysis technique*

# Results

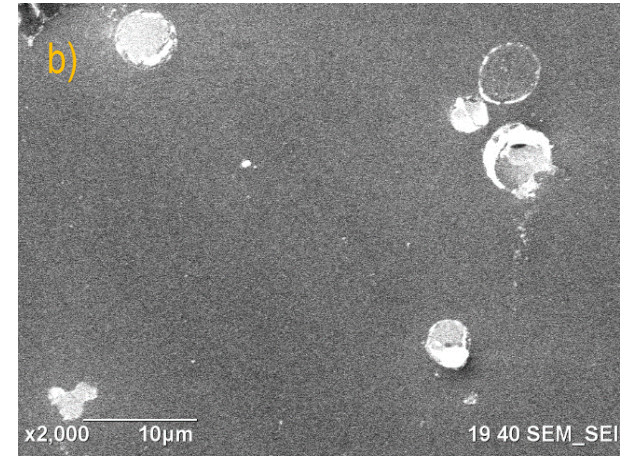
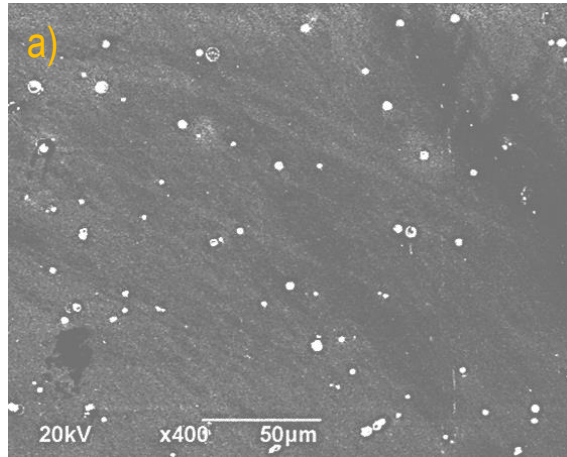
In order to properly analyze the thickness and morphology of the film, Graph 1 shows the thickness profile obtained by profilometry, where column-type microstructured arrangements with an average height of 805 nm can be observed, the separation among these growths has micron variations where the growth of a thin film of BFO with a thickness of approximately 150 nm is also observed.



**Graph 1** Thickness profile and average thickness for the BFO film deposited by Spray Pyrolysis.

# Results

Figure 2 present the SEM micrographs obtained from the film, where subsection a) shows the presence of the columns observed by profilometry and their distribution fully coincides with the profile obtained from the sample, also in part b) it is shown a magnification x2000 where these columns have a circular shape with a variable diameter, but with an average of 5  $\mu\text{m}$ .

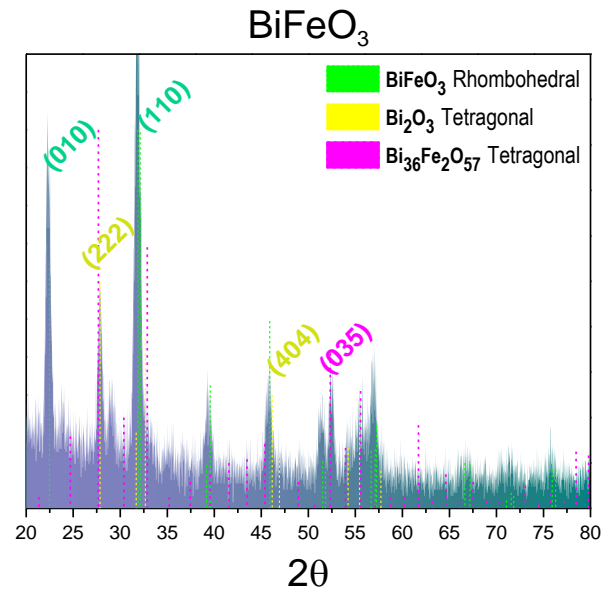


**Figure 2** Micrographs of the BFO film obtained by SEM with different magnifications.

# Results

Graph 2 shows the diffraction pattern, where we present the rhombohedral phase of  $\text{BiFeO}_3$  with a preferential orientation in the (110) plane, it is also observed that there is a mixture with the tetragonal phase of  $\text{Bi}_2\text{O}_3$  which presents a preferential orientation in the plane (222).

Both phases present in the film correspond to the structure at room temperature for each case. Finally, the presence of the tetragonal phase of  $\text{Bi}_{36}\text{Fe}_2\text{O}_{57}$  with a preferential orientation in the (035) plane is also observed to a lesser extent.



**Graph 2** Diffractogram of BFO deposited by Spray Pyrolysis and diffraction patterns of rhombohedral  $\text{BiFeO}_3$ , tetragonal  $\text{Bi}_2\text{O}_3$  and tetragonal  $\text{Bi}_{36}\text{Fe}_2\text{O}_{57}$ .

# Results

Figure 3 shows a structural representation of the phases and materials obtained using the lattice parameters found in the diffraction pattern of the film obtained by spray pyrolysis.

The lattice parameters are as follows;

**BiFeO<sub>3</sub>** a= 3.952 Å, b= 3.952 Å, c= 3.952 Å y  $\alpha=\beta=\gamma= 89.6^\circ$ .

**B<sub>2</sub>O<sub>3</sub>** a= 7.7299 Å, b= 7.7299 Å, c= 5.6587 Å y  $\alpha=\beta=\gamma= 90^\circ$ .

**Bi<sub>36</sub>Fe<sub>2</sub>O<sub>57</sub>** a= 10.184 Å, b= 10.184 Å, c= 10.184 Å y  $\alpha=\beta=\gamma= 90^\circ$ .

from the diffraction pattern, mean crystallite sizes are calculated using the Scherrer's formula from the broadening of the peaks.

Scherrer's formula is:

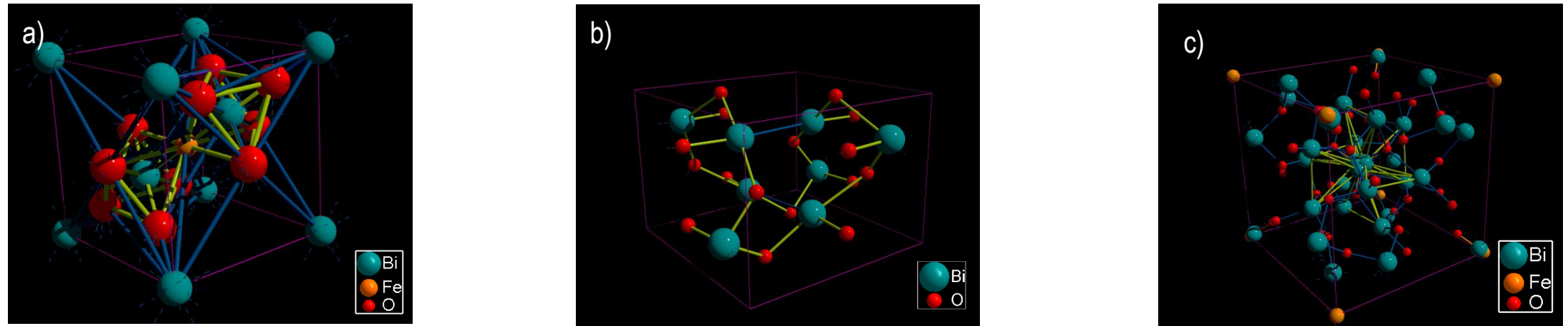
$$t = \frac{K\lambda}{\beta \cos \theta} \quad (1)$$

where t is the crystallite size, K is Scherrer's constant (0.94),  $\lambda$  is the wavelength of the radiation source (1.59 Å),  $\beta$  is the full width at half maximum, and  $\theta$  is the angle of Bragg [10].

The results obtained show an average crystallite diameter of approximately 13 nm for BFO, as well as for Bi<sub>2</sub>O<sub>3</sub>.



# Results

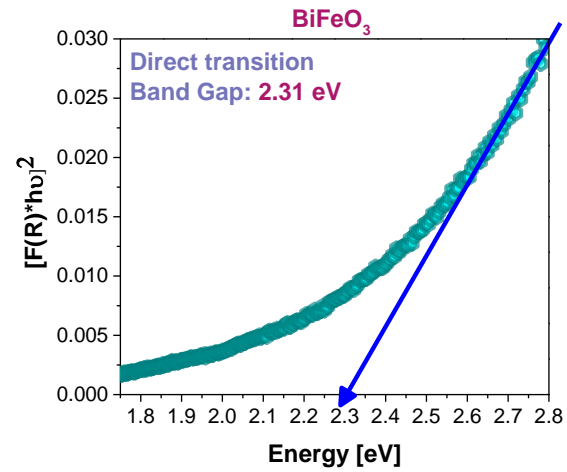


**Figure 3** Representation of the different phases present in the obtained film a)  $\text{BiFeO}_3$  rhombohedral, b)  $\text{Bi}_2\text{O}_3$  tetragonal, c)  $\text{Bi}_{36}\text{Fe}_2\text{O}_{57}$  tetragonal.

Regarding to the optical properties of the material, Graph 3 shows the band gap calculated by means of diffuse reflectance measurements and the use of the Kubelka-Munk function [11] where a direct type transition was used in accordance with what was reported for the BFO [12].

The band gap obtained corresponds to 2.31 eV, so there is a shift in the band gap of the material towards lower energies.

# Results



**Graph 3** Band gap of the BFO film calculated by the Kubelka-Munk function using a direct transition.

# Analysis

From the X-ray diffractograms it is observed as mentioned above, that in the film deposit we obtain the rhombohedral phase of  $\text{BiFeO}_3$  in combination with the tetragonal phase of  $\text{Bi}_2\text{O}_3$  and to a lesser extent the  $\text{Bi}_{36}\text{Fe}_2\text{O}_{57}$  in tetragonal phase. Spectra such as the ones obtained have been reported in investigations where the union of  $\text{BiFeO}_3$ - $\text{Bi}_2\text{O}_3$  films is analyzed [13], as well as the heterojunction of nanofibers and other composites of these 2 materials [14]. These results, together with those obtained by SEM, could indicate that a  $\text{BiFeO}_3$  film is actually being obtained, so  $\text{BiFeO}_3$  is embedded with  $\text{Bi}_2\text{O}_3$  (clusters or columns observed in figure 2), although according to the size of the crystallites, this mixture of phases could be considered to be at the nanometric level with nanocrystalline embedding. This result is very interesting from the magnetic point of view since it has been reported [12] that the magnetic properties in BFO increase as the crystallite size decreases.

Regarding the  $\text{Bi}_{36}\text{Fe}_2\text{O}_{57}$  phase, it is reported that small amounts of bismuth-rich secondary phases such as  $\text{Bi}_{40}\text{Fe}_2\text{O}_{63}$  and  $\text{Bi}_{36}\text{Fe}_2\text{O}_{57}$  are commonly found in polycrystalline samples of  $\text{BiFeO}_3$  and the segregation of these secondary phases at the grain boundaries could play a role in reducing the resistivity of the material. The low resistivity, and resulting low bias inhibit the ferroelectric hysteresis loop [15].

The  $\text{Bi}_{36}\text{Fe}_2\text{O}_{57}$  phase, as well as the  $\text{Bi}_2\text{O}_3$  phase, indicate an excess of bismuth in our film, which could be explained in terms of the low deposition temperature of  $100^\circ\text{C}$  used and the heat annealing carried out at  $500^\circ\text{C}$ .

The  $\text{Bi}_2\text{O}_3$  is a p-type bismuth-based semiconductor with unique physical and chemical characteristics which make it promising in photoelectric applications, it has a band gap of 2.2 to 2.8 eV and shows good absorption in the visible region of the light [14].

According to the results published by [16],  $\text{BiFeO}_3$  is a typical n-type semiconductor, so the heterojunction of  $\text{Bi}_2\text{O}_3/\text{BiFeO}_3$  forms a p-n junction type compound, thus the photogenerated charges are effectively separated and transferred at the interface of  $\text{Bi}_2\text{O}_3$  and  $\text{BiFeO}_3$  [14].

# Analysis

Furthermore, a high photovoltaic performance in  $\text{BiFeO}_3$  films by manipulating the concentration of their oxygen vacancies through the alteration of the Bi content has been reported, the results of this research show that the highest photovoltaic production was achieved in  $\text{Bi}_{1.05}\text{FeO}_3$  films, whose response is 1000 times better than that of  $\text{Bi}_{0.95}\text{FeO}_3$  films, as a charge balance requirement in BFO, Bi vacancies lead to an increase in the oxygen vacancy concentration, while an excess of Bi is linked to a decrease in the oxygen vacancy concentration [17].

Analyzing the shift of the band gap obtained towards lower energies, this can be attributed to the mixture of phases with the insertion of  $\text{Bi}_2\text{O}_3$  in the BFO molecular structure present in the film, which again corroborates what is observed from the diffractogram of Graph 2, this shift is also beneficial for light absorption according to what was reported in [13] where the decrease in the band gap is obtained by doping the BFO, so in our case the presence of important quantities of  $\text{Bi}_2\text{O}_3$  molecules causes the reduction of the bandgap energy of the film.

In this context, it is easy to discern that obtaining the reported  $\text{BiFeO}_3\text{-Bi}_2\text{O}_3$  film represents an enormous advantage in terms of materials that are to be applied in the field of photovoltaics (both in solar cells and photocatalytic applications), since it is a film with both phases well defined, besides to the fact that the deposit technique is simple and economical when using the pyrolysis spray system.

# Conclusions

In the present research, the obtaining of  $\text{BiFeO}_3\text{-Bi}_2\text{O}_3$  films by means of the ultrasonic Spray Pyrolysis technique is reported, with the formation of 2 phases in the material, namely,  $\text{BiFeO}_3$  with a rhombohedral structure and  $\text{Bi}_2\text{O}_3$  with a tetragonal phase, additionally to the presence of the tetragonal  $\text{Bi}_{36}\text{Fe}_2\text{O}_{57}$  phase, this is indicative of an excess bismuth in the material possibly attributed to the low deposition temperature. Obtaining this type of structures by the ultrasonic Spray Pyrolysis technique is so important for future photoelectric applications due to the various investigations are focused on obtaining this type of arrangement by sophisticated techniques. The reported structures possess an excellent response to light, such fact makes them good candidates to improving photovoltaic cells based on multiferroic materials.

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